First-principles study on competing phases of silicene: Effect of substrate and strain

Chi-Cheng Lee (李敬正),1 Antoine Fleurence,1 Rainer Friedlein,1 Yukiko Yamada-Takamura,1 and Taisuke Ozaki1,2
1School of Materials Science, Japan Advanced Institute of Science and Technology (JAIST), 1-1 Asahidai, Nomi, Ishikawa 923-1292, Japan
2Research Center for Simulation Science, Japan Advanced Institute of Science and Technology (JAIST), 1-1 Asahidai, Nomi, Ishikawa 923-1292, Japan

(Received 3 July 2013; revised manuscript received 23 August 2013; published 4 October 2013)

The stability and electronic structure of competing silicene phases under in-plane compressive stress, either free-standing or on the ZrB2(0001) surface, has been studied by first-principles calculations. A particular (\(\sqrt{3} \times \sqrt{3}\))-reconstructed structural modification was found to be stable on the ZrB2(0001) surface under epitaxial conditions. In contrast to the planar and buckled forms of free-standing silicene, in this “planar-like” phase, all but one of the Si atoms per hexagon reside in a single plane. While without substrate, for a wide range of strain, this phase is energetically less favorable than the buckled one, it is calculated to represent the ground state on the ZrB2(0001) surface. The atomic positions are found to be determined by the interactions with the nearest-neighbor Zr atoms competing with Si-Si bonding interactions provided by the constraint of the honeycomb lattice.

DOI: 10.1103/PhysRevB.88.165404 PACS number(s): 73.22.−f, 68.43.Fg, 73.20.−r

I. INTRODUCTION

Owing to its ability to be either \(sp^3\)- or \(sp^2\)-hybridized, under ambient conditions, carbon crystallizes in different forms, such as diamond, graphite, graphene, fullerenes, and carbon nanotubes. Such diversity does, of course, inspire the search for similar graphitic forms of the element silicon, which is just below carbon in the periodic table. Whereas silicon nanotubes have already been reported,1,2 until recently, the two- and three-dimensional graphitic forms of silicon remained elusive but have been anticipated theoretically. First-principles studies on the stability and the properties of graphitic Si phases can be traced back decades ago3 and have mostly focused on an atom-thick honeycomb lattice of Si atoms, called silicene, as the Si-counterpart of graphene. In contrast to graphene, however, it is well accepted that silicene is unstable as long as the structure is planar.3–5 Instead, free-standing silicene is predicted to be stable in a so-called “buckled” structure, where the two sublattices of the bipartite lattice are at different heights.5 Like in graphene, charge carriers in free-standing silicene are found to be massless Dirac fermions even if it is buckled.4,5 The Fermi velocity is calculated to be in the order of \(10^6\) m/s.3 Additionally, due to a spin-orbit coupling much stronger than that in graphene, the quantum Hall effect is predicted to occur4,6 and, owing to its buckled structure, to be switchable by the application of a perpendicular electric field.7 Since the many fascinating properties and countless potential applications proposed for graphene8–18 may occur in silicene as well, it is of great importance to uncover all stable structural modifications of silicene.

So far, Si honeycomb structures have been reported to grow on a limited number of substrates, such as Er, Ag, ZrB2, and Ir.19–29 DFT calculations in conjunction with experimental observations point out that all of those two-dimensional forms of silicon are deviating from the expected buckled form of free-standing silicene.21–29 Among them, those on Ag(111) and on ZrB2(0001) surfaces are the most exciting ones since the experimental observation of \(\pi\) bands links them to the theoretical concept of silicene. Imposed by epitaxial conditions, the type and degree of buckling can vary by a large amount, which in turn influences the electronic properties as well. The intimate relationship between the structure and the electronic properties makes the atomic-scale buckling a highly relevant parameter in the description of silicene. On the other hand, the structural flexibility, related to a mixed \(sp^2/sp^3\) hybridization, may allow the engineering of desired properties such as the opening of a gap, something that is difficult to achieve in purely \(sp^2\)-hybridized, robust graphene.28

For silicene on the Ag(111) surface, the growth behavior is strongly dependent on the temperature, the coverage, and the deposition rate.21–26 Among the reported superstructures, postannealing cannot switch one observed phase to another,23 which advocates the presence of nonnegligible energy barriers between these phases. It has been proposed that all the observed silicene superstructures exhibit different orientations with respect to the silver substrate, which is manifested in particular surface reconstructions.27 On the contrary, epitaxial silicene on zirconium diboride (0001), formed by the segregation of atoms from the substrate on the surface of ZrB2(0001) thin films grown on Si(111) wafers,28 has the advantage to exhibit a single orientation, to cover the whole sample surface homogeneously, and to be very well reproducible. The latter is a direct consequence of the spontaneous and self-terminating growth mode. Epitaxial silicene on ZrB2(0001) is \((\sqrt{3} \times \sqrt{3})\)-reconstructed, which is provided by the commensurate relationship of the \((\sqrt{3} \times \sqrt{3})\) unit cell of silicene with the \((2 \times 2)\) unit cell of ZrB2(0001).28 The spontaneous formation of silicene on ZrB2(0001) suggests that this particular reconstruction is important for the stability of silicene. The nature of this \((\sqrt{3} \times \sqrt{3})\) reconstruction has recently been discussed based on density functional theory (DFT) calculations.28 Note, however, that the considered phase, whose calculated structural and electronic properties are in partial agreement with the experimental findings,28,30 has been calculated to be a metastable structure. It deviates
from that of the predicted free-standing silicene by the position of
one of the atoms forming the bottom sublattice lifted up to
almost the height of the top sublattice. 28 On the other hand,
the properties of the ground state that are discussed in the
present paper for the first time do not fit with the experimental
results. In the ground state proposed by DFT, the structure
is “planar-like,” closer to that of planar graphene than to the
metastable phase. In this structure, five out of the six Si atoms
per hexagon reside in a plane characterized by a residual
buckling of just 0.01 Å, with the remaining atom protruding
at a height of 1.58 Å.

Note that as a similar (√3 × √3)-reconstructed planar-
like phase has been calculated to be one of the silicene structures
forming on the Ag(111) surface, 27 it is of great
interest to investigate under which conditions this particularly
reconstructed phase may become stable. Here, interactions
with the substrate and epitaxial strain are important issues.
In this paper, we report the results of our theoretical study
describing the competition between structural variations of
(√3 × √3)-reconstructed silicene under the influence of the
ZrB2(0001) substrate and in-plane compressive strain. The
discussion sheds light onto both the structural parameters and
the role of the substrate electronic properties on the stability
of epitaxial silicene. The paper is organized as follow: After
presenting the details of the computational methods (Sec. II),
we will discuss the stability of the different silicene structures
in a stepwise manner. First, free-standing silicene (Sec. III)
and the ZrB2(0001) substrate itself (Sec. IV) are considered. In
Secs. V and VI are presented the interactions of single Si atoms
and of Si honeycomb layers in contact with the ZrB2(0001)
substrate, respectively. Discussion and conclusions can be
found in Sec. VII.

II. COMPUTATIONAL DETAIL
DFT calculations within a generalized gradient approximation 29,30
have been performed using the OpenMX
code, which is based on norm-conserving pseudopotentials
generated with multireference energies 31 and optimized
pseudoatomic basis functions. 32,33 The regular mesh of
270 Ry in real space was used for the numerical integrations
and for the solution of the Poisson equation. The cut-off
radius of 7 Bohr has been chosen for all the basis functions.
For each Zr atom, three, three, and two optimized radial
functions have been allocated for the
sp, pσ, and d-orbitals, respectively, as denoted by Zr-3p3d2. For B and Si atoms,
B-4s2p2d1 and Si-2p2d1 configurations have been adopted.
The spin-orbit coupling has not been considered in our
calculation. The calculated lattice constants of bulk ZrB2,
a = 3.17 Å and c = 3.55 Å, obtained using a 8 × 8 × 5
mesh of k points in reciprocal space are in good agreement
with experimental data. 34 For the study of free-standing silicene,
of the ZrB2(0001) surface and of silicene on the
ZrB2(0001) surface, the in-plane lattice constants of the
bulk and a 4 × 4 × 1 mesh of k points have been used
for the unit cell corresponding to the (2 × 2)-reconstructed
ZrB2(0001) surface. For the calculation of surface properties,
the structural parameters of the central two Zr and three B
layers of a 15-layer Zr-terminated slab have been fixed to
the bulk values. Other atomic positions have been relaxed
until the residual force on each atom has reached values
of less than 3 × 10−4 Hartree/Bohr. In order to allow for
a symmetric arrangement, silicene layers containing six Si
atoms per (2 × 2) unit cell of the ZrB2(0001) surface have
been placed on both sides of the Zr-terminated slab.

In terms of the representation of the electronic band
structures, the Brillouin zones of the (1 × 1) unit cell
of silicene and the (1 × 1 × 1) unit cell of bulk ZrB2 have
been chosen for the studies of free-standing silicene and ZrB2(0001)
surface, respectively. For the case of silicene on the ZrB2(0001)
surface, both Brillouin zones have been chosen. In the study
of silicene-derived bands, the two symmetric silicene layers
on each surface of the slab are considered as two separate
layers. Since the supercell is larger than the primitive unit
unit cell chosen for the representation, an additional procedure 35
will unfold the bands into larger zones is needed and has
been generalized to the nonorthogonal pseudoatomic basis
functions. 36 In order to show the spectral weight related to each
eigenstate of the system at a k point, we used the so-called “fat
band representation,” where the weight was represented by
the diameter of a circle. The respective contributions from
different orbitals to the spectral weight are given by the
diameters of color-coded circles, as specified in the figure
captions. The diameter of gray-colored circle is obtained by
summing over all the orbital contributions of an eigenstate. For
a better visualization, eigenstates with the same k and separated
by an energy differences less than 0.02 eV are considered as
being degenerated and represented by a single circle.

In order to assure structural convergence of the ground
state of silicene on the ZrB2(0001) surface calculated by
DFT, additionally, we have performed molecular dynamics
calculations starting from different initial structures, the
temperature set to 400 K. The same ground-state structure
has been found in all of the molecular dynamics simulations.
A consistent ground-state structure of silicene on the (2 × 2)
unit cell of the ZrB2(0001) surface was also found by using
the VASP package with ultrasoft pseudopotentials. 37,38 To analyze
the energy barrier between the ground state and a proposed
metastable phase, 16 transition images are adopted using the
nudged elastic band (NEB) method where the minimum energy
path between two configurations can be found. 41 Here, the
maximum force is less than 5 × 10−3 Hartree/Bohr.

III. FREE-STANDING SILICENE
Determined by the sp2 hybridization similar to graphene,
free-standing planar silicene is expected to display comparable
electronic properties. 5 As shown in Fig. 1(a), several hopping
parameters can contribute to the electronic band structure
of planar silicene. This can well be explained within a
tight-binding approach considered here initially. A finite set of hopping integrals related to the σ-bond hopping between
two of the sp2 orbitals (t1), the pσ − pσ hopping (t2), and the
dominant pσ − sp2 hopping (t3) shall be used. In reciprocal
space, the Hamiltonian, 41

where a∗ and b∗ denote the reciprocal lattice vectors
of the unit cell containing two Si atoms. As a consequence
of the given symmetry, this Hamiltonian describes several
interesting properties of the planar structure of silicene, which
are discussed below.
Because of the planarity, $t_3$ is zero. Therefore, $H(\vec{k})$ in the $p_z$ subspace is decoupled from that of the other orbitals. As a consequence, this simply leads to the presence of a bonding ($\pi$) band and an antibonding ($\pi^*$) band. The off-diagonal term $(t_2 + t_2 e^{i 2\pi k_x} + t_2 e^{-i 2\pi k_x})$ becomes zero at $\vec{k} = \vec{K} \equiv (\frac{1}{2}, \frac{1}{2})$, forming a Dirac point, and a linear dispersion can be derived as $E(\vec{k}) = \epsilon_{p_z} \pm \sqrt{3} |t_2| a q / a z$ in the vicinity of the $K$ point, where $q = \vec{q} - \vec{K}$. This is illustrated in Fig. 1(b). In the half-filled case, the Fermi energy is equal to the site energy $\epsilon_{p_z}$ and thus coincides with the energy of the Dirac point. The simple analysis reveals the unique signature of the $p_z$-derived bands formed in the honeycomb lattice. On the other hand, within this model, the $sp^2$ orbitals form flat bands at energies of $\epsilon_{sp^2} \pm |t_1|$.

$$
\begin{pmatrix}
\epsilon_{sp^2} & 0 & 0 & 0 & 0 & 0 & t_1 & 0 & 0 & 0 \\
0 & \epsilon_{sp^2} & 0 & 0 & 0 & 0 & 0 & t_1 e^{i 2\pi k_y} & 0 & 0 \\
0 & 0 & \epsilon_{sp^2} & 0 & 0 & 0 & -t_3 & 0 & 0 & 0 \\
0 & 0 & 0 & \epsilon_{p_z} & 0 & 0 & \epsilon_{p_z} & 0 & 0 & 0 \\
0 & 0 & 0 & t_1 e^{i 2\pi k_y} & 0 & -t_1 e^{-i 2\pi k_y} & t_3 + t_3 e^{i 2\pi k_y} + t_3 e^{-i 2\pi k_y} & 0 & 0 & 0 \\
0 & 0 & 0 & t_1 e^{-i 2\pi k_y} & 0 & -t_1 e^{i 2\pi k_y} & t_3 e^{i 2\pi k_y} + t_3 e^{-i 2\pi k_y} & 0 & 0 & 0 \\
0 & 0 & 0 & t_2 e^{i 2\pi k_x} + t_2 e^{-i 2\pi k_x} & 0 & -t_2 e^{i 2\pi k_y} - t_2 e^{-i 2\pi k_y} & 0 & 0 & 0 & 0 \\
0 & 0 & 0 & t_2 e^{-i 2\pi k_x} + t_2 e^{i 2\pi k_x} & 0 & -t_2 e^{-i 2\pi k_y} - t_2 e^{i 2\pi k_y} & 0 & 0 & 0 & 0 \\
\end{pmatrix}
$$

(1)

Another interesting finding revealed by the analysis of $H(\vec{k})$ is related to the bonding state characterized by $E = \epsilon_{sp^2} - |t_1|$ and the corresponding antibonding state with $E = \epsilon_{sp^2} + |t_1|$. These states are preserved even in the case of a nonzero $t_3$ and carry no $p_z$ component. The energies of the two states are immune to the buckling. Within the whole Brillouin zone, the corresponding eigenstates involve only $sp^2$ orbitals. Unlike these two flat bands, all the other bands derive from a mixture of $sp^2$ and $p_z$ orbitals. The three bonding bands and three antibonding bands defined in the $p_z$ subspace are illustrated in Fig. 1(b). The buckling gives rise to an additional modification in these electronic states manifested in an avoided crossing of the bonding-type (or antibonding-type) bands as shown in Fig. 1(c).

In the framework of the DFT calculations, in addition to the planar and buckled structures of silicene, we considered the $(\sqrt{3} \times \sqrt{3})$-reconstructed structure as well since it has been predicted to be relevant for epitaxial systems.\textsuperscript{27,28} For comparison, we also evaluated the planar and buckled forms within a supercell in the size of the primitive unit cell of this “planar-like” structure. Hereafter, the buckled phase will be referred to as the “regularly buckled” one in order to distinguish the two nonplanar phases. In Fig. 2, the geometrical structures of the planar, the regularly buckled, and the planar-like phases are plotted together with their total energies as a function of the in-plane lattice constant $a$ of the $(\sqrt{3} \times \sqrt{3})$ unit cell.

With one Si atom per unit cell protruding out of plane, the planar-like phase is able to sustain a longer in-plane bond length as compared to the planar phase such that it becomes closer to that of the regularly buckled one. As shown in Fig. 2, in regularly buckled silicene, the buckling height of 0.69 Å at $a = 6.35$ Å decreases to 0.50 Å at $a = 6.70$ Å, which is a

![FIG. 1. (Color online) (a) Nearest-neighbor-hopping integrals within a simple tight-binding model for a Si honeycomb lattice considering only the $\sigma$-bond hopping ($t_1$), $p_z$ -- $p_z$ hopping ($t_2$), and the dominant $p_z$ -- $sp^2$ hopping ($t_3$). $t_3$ is zero for the planar structure but can reflect the strength of buckling between the two atoms in the unit cell. A richer dispersion in the $sp^2$ branches can further be introduced by including other hopping terms, by for example, $t_4$, $t_5$, and $t_6$. (b) Scheme of the band structure revealed by applying the Hamiltonian in Eq. (1) to the $t_1$ and $t_2$ terms alone. Horizontal lines represent $\sigma$ bands and the X-shape lines illustrate the $\pi$ bands forming a Dirac-cone structure at $K$. The bonding and antibonding characters of bands in the $p_z$ subspace are indicated by green and orange colors, respectively. The black dashed lines show bands without $p_z$ character. (c) Schematic band structure with nonzero $t_3$.](image)
The bond lengths and the buckling heights of the three phases are indicated for the particular lattice constant of the \( \sqrt[3]{3} \times \sqrt[3]{3} \) unit cell of silicene. The bond lengths and the buckling heights of the three phases are indicated for the particular lattice constant of \( a = 6.35 \) Å, which corresponds to that of the \( (2 \times 2) \) unit cell of the ZrB\(_2\)(0001) surface. This particular value of the lattice constant is indicated by the dashed line. Upon increase of \( a \) to more than 6.60 Å, the planar-like structure relaxes to the planar structure.

Since the total energies of regularly buckled and planar-like phases are generally lower than that of the planar one, it is important to verify again if the energy gain might be of electronic origin or not. For this purpose, we consider the DFT results for \( a = 6.35 \) Å, which is the in-plane lattice constant of the \( (2 \times 2) \) unit cell of ZrB\(_2\)(0001). The electronic structures of these three phases represented in the first Brillouin zone of the silicene \((1 \times 1)\) unit cell are shown in Fig. 3. As expected, the band structures of the planar and regularly buckled phases are quite similar and do not show any additional symmetry breaking introduced by the \( \sqrt[3]{3} \times \sqrt[3]{3} \) supercell. In Figs. 3(a) and 3(b), occupied \( \pi \) and unoccupied \( \pi^* \) bands with \( p_z \) character are colored in black. At the \( K \) point, a Dirac cone is found at \( E_F \), which is in agreement with our analysis using the simple tight-binding model and earlier studies.\(^{4,5}\)

While for planar silicene, the \( p_z \) orbitals do not couple to the \( sp^2 \) orbitals, the buckling leads to nonzero hopping integrals between the \( p_z \) and the neighboring \( sp^2 \) orbitals, which confer a mixed \( \pi - \sigma \) character to the originally pure \( \pi \) and \( \pi^* \) bands. Unlike one of the \( \sigma \) band marked in green in Figs. 3(a) and 3(b) that carries negligible \( p_z \) weight, the other occupied \( \sigma \) band mixes with the bonding \( \pi \) band such that band crossing is avoided. Although the resulting gap is large, the corresponding bonding and antibonding bands are both occupied such that no band energy is gained. Similarly, the energy of the \( \sigma \) band at the \( \Gamma \) point is increased by more than 1 eV. Therefore, the mechanism that stabilizes the regularly buckled silicene over the planar one is hard to visualize in the electronic band structure itself. It may instead relate to an instability in the phonon part that involves the lattice repulsive potential and the response of electrons to the lattice vibration.\(^{4,5}\)

The electronic structure of planar-like silicene is presented in Fig. 3(c). A strong breaking of the symmetry causes back-folding and the lifting of the degeneracy of bands. Despite the back-folding, some resemblance to planar silicene is found. In particular, the energies of the \( \sigma \) band at the \( \Gamma \) point are similar. If the origin of the stabilization of the planar-like phase with respect to the planar one in terms of the total energy is not found in the band structure, other energy terms shall be considered. And indeed, it is found that energy is gained mainly from a reduction of the core-core Coulomb repulsion of the nuclei. Quantitatively, this repulsion is lowered by about 50 and 60 eV per Si atom, as compared to the regularly buckled and planar phases, respectively. This suggests an essential role of the atomic positions and the bond lengths for the stability.
of the planar-like phase that might provide a competitive advantage over other phases under restraints imposed by epitaxial conditions on substrates. Note that cone-like band dispersions are still noticeable even under the \((\sqrt{3} \times \sqrt{3})\) reconstruction, which, however, leads to the opening of a small gap and an up-shift of the corresponding features. A similar dispersion was also found in the band structure calculated for freestanding silicene having the structure of epitaxial silicene on Ag(111) substrate.27 This can be visualized at the \(K\) point via unfolding, as shown in Fig. 3.

Considering again a tight-binding approach using Wannier functions,42,43 the deviation of the band structure of the planar-like phase from those of both the planar and regularly buckled phases can be explained by the breaking of both the threefold rotation and inversion symmetries. This is also expressed in the splitting of on-site energies and hopping integrals provided in Table I. Note that for the planar-like phase, not only differences related to the indicated vibration, for example, in the presence of a substrate. With the presentation of the total energy as a function of the amount of displacements, for the case of planar-like silicene relaxed onto the surface of ZrB\(_2\)(0001), a stable Born-Oppenheimer surface can be found. The dramatic change of the total energy surface implies that the substrate does play a crucial role in the stabilization of epitaxial silicene structures.

\section{IV. The Zirconium Diboride (0001) Surface}

Before going to epitaxial silicene on the diboride substrate, the electronic structure of the bare Zr-terminated ZrB\(_2\)(0001) surface shall be discussed. Electronic structure calculations with results similar to those published previously have been performed for the bulk and for a slab, the model of which is shown in Fig. 5(a). Electronic states related to contributions from orbitals of the bulk and of the outermost ZrB\(_2\) layer of the slab are presented in Figs. 5(b) and 5(c), respectively. The Zr \(d\) character of bands is color-coded in red. Clearly, for the outermost layer, several bands are observed that do not have a counterpart in the electronic structure of the bulk. The most prominent surface state has a dominant Zr \(d\) character and crosses \(E_F\) twice between the \(\Gamma\) and \(M\) points. Experimentally, an electronic feature assigned to this surface state has been found to strongly hybridize with silicene-derived electronic states upon donation of negative charge from adsorbed potassium atoms, and may be crucial for silicene-substrate interactions even in the pristine case.

\section{V. The Zirconium Diboride (0001) Surface with Single Adsorbed Silicon Atoms}

For the study of the adsorption of Si atoms on the ZrB\(_2\)(0001) surface, the \((2 \times 2)\) unit cell of ZrB\(_2\)(0001) is considered. This unit cell has been chosen because of its commensurate relationship with the \(\sqrt{3} \times \sqrt{3}\) unit cell of reconstructed silicene. As shown in Figs. 6(a)–6(c), three possible symmetric sites—the on-top, hollow, and bridge
FIG. 5. (Color online) (a) Structure of the slab used in the calculations of properties of the $(2 \times 2)$ unit cell of the Zr-terminated ZrB$_2$(0001) surface. The central five layers have been fixed to the bulk structure. The respective electronic band structures of (b) bulk ZrB$_2$ and (c) the slab, both represented within the first Brillouin zone of bulk ZrB$_2$. The Zr $d$ and B $p$ characters of bands are colored in red and blue, respectively. Only the contribution of the outermost ZrB$_2$ layers of the slab are shown in the fat-band representation. Newly revealed bands in the vicinity of the Fermi energy are identified by their dispersions as the surface states.

For all positions, the Zr atoms adjust their heights as well, such that those closest to the added Si atom adopt a slightly lower position with respect to the remaining Zr atoms in the terminating layer. The out-of-plane coordinates are listed in Table II.

From the information thus obtained, favorable in-plane positions for the Si atoms forming silicene on the ZrB$_2$(0001) surface can be predicted. The best candidate would be a structure in which all of the Si atoms reside at hollow sites. This is, however, impossible to realize for six Si atoms per $(2 \times 2)$ unit cell since the Si honeycomb lattice prefers a larger lattice constant than the boron sublattice of ZrB$_2$. On-top sites could also be avoided by locating all six Si atoms in near-bridge sites, that is in positions in between on-top and bridge sites. However, the deviation from the perfect bridge position would increase the energy too much such that it is preferable to place two Si atoms at hollow sites, three Si atoms at near-bridge sites, and one Si atom at an on-top site. This simple analysis provides

TABLE II. The heights of outermost Zr atoms [cf. Fig. 6] measured from the second outermost Zr layer for a single Si atom adsorption on ZrB$_2$(0001) surface. The values without Si atom adsorption are also shown as comparison. The unit is Å.

<table>
<thead>
<tr>
<th>Si adsorption site</th>
<th>Zr atom</th>
<th>No Si</th>
<th>On-top</th>
<th>Hollow</th>
<th>Bridge</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>I</td>
<td>3.48</td>
<td>3.54</td>
<td>3.56</td>
<td>3.59</td>
</tr>
<tr>
<td></td>
<td>II</td>
<td>3.48</td>
<td>3.54</td>
<td>3.50</td>
<td>3.59</td>
</tr>
<tr>
<td></td>
<td>III</td>
<td>3.48</td>
<td>3.54</td>
<td>3.50</td>
<td>3.46</td>
</tr>
<tr>
<td></td>
<td>IV</td>
<td>3.48</td>
<td>3.37</td>
<td>3.50</td>
<td>3.46</td>
</tr>
</tbody>
</table>

For all positions—allow Si atoms to avoid in-plane forces. The positions have been relaxed along the out-of-plane direction. For each position, the corresponding heights as given with respect to the uppermost Zr atom and the binding energies are defined as $(E[\text{ZrB}_2 \text{ slab}]+E[\text{two Si atoms}]-E[\text{ZrB}_2 \text{ slab + two Si atoms}])/2$ and indicated in Fig. 6 as well. It is found that the on-top position is the least favorable one. As understandable from the corresponding coordination numbers, Si atoms prefer the hollow and bridge sites, which brings them closer to the Zr layer. Obviously, the on-top Si atom that is furthest away from the Zr layer can bind only to a single Zr atom and is thus more weakly interacting which is different from the hollow and bridge sites where three and two Zr atoms are affected, respectively.

FIG. 6. (Color online) Individual Si atoms at (a) the on-top, (b) the hollow, and (c) the bridge positions plotted for one Si atom per $(2 \times 2)$ unit cell of the ZrB$_2$(0001) surface. The corresponding binding energy per Si atom ($E_{\text{binding}}$) and the height ($h$) to its nearest Zr atom are shown as well. The distance to its nearest Zr atom can be found as (a) 2.57 Å, (b) 2.68 Å, and (c) 2.61 Å.

TABLE II. The heights of outermost Zr atoms [cf. Fig. 6] measured from the second outermost Zr layer for a single Si atom adsorption on ZrB$_2$(0001) surface. The values without Si atom adsorption are also shown as comparison. The unit is Å.

<table>
<thead>
<tr>
<th>Si adsorption site</th>
<th>Zr atom</th>
<th>No Si</th>
<th>On-top</th>
<th>Hollow</th>
<th>Bridge</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>I</td>
<td>3.48</td>
<td>3.54</td>
<td>3.56</td>
<td>3.59</td>
</tr>
<tr>
<td></td>
<td>II</td>
<td>3.48</td>
<td>3.54</td>
<td>3.50</td>
<td>3.59</td>
</tr>
<tr>
<td></td>
<td>III</td>
<td>3.48</td>
<td>3.54</td>
<td>3.50</td>
<td>3.46</td>
</tr>
<tr>
<td></td>
<td>IV</td>
<td>3.48</td>
<td>3.37</td>
<td>3.50</td>
<td>3.46</td>
</tr>
</tbody>
</table>

165404-6
the in-plane structure model of silicene that has independently been derived from experimental data.28

If we assume that Si atoms would adopt the height calculated for individual Si atoms as listed in Figs. 6(a) to 6(c), the planar-like silicene structure can easily be accommodated on the ZrB$_2$(0001) surface. With the protruding Si atom sitting at an on-top position, the other five Si atoms can be located at hollow sites and near-bridge sites, for two and three Si atoms, respectively. On the contrary, for the regularly buckled silicene structure, the height profile does not follow the favorable positions found by this simple analysis since the Si atoms sitting on hollow and on-top sites belong to the same sublattice. With the restrain of limiting the surface density to exactly six atoms per ($2 \times 2$) unit cell of the ZrB$_2$(0001), the proposed planar-like structure of silicene is indeed calculated to be the ground state, as shown in Sec. VI. This points to the dominant role of the local environment of individual silicon atoms for the structural properties of epitaxial silicene.

Note that with approximately 5 eV, the binding energy per Si atom is quite large. Due to the symmetry breaking caused by the added Si atom and the interactions involved, for all of the three adsorption sites, the Zr-$d$-derived surface state splits into bonding and antibonding states of which the former is shifted down in energy and the latter becomes unoccupied (band structures not shown). With hollow sites showing the strongest interactions, the corresponding Si $p_z$ orbitals hybridize most strongly with Zr $d$ orbitals of the outermost layer.

VI. EPITAXIAL SILICENE ON THE ZIRCONIUM DIBORIDE (0001) SURFACE

In this section, the results for six Si atoms per ($2 \times 2$)-reconstructed ZrB$_2$(0001) unit cell are reported. In the search for possible silicene structures beyond the one predicted in Sec. V, only two minima of the total energy have been found. The associated two phases, shown in Figs. 7(a) and 7(b), resemble structurally free-standing ones, which are the regularly buckled and planar-like phases, considered in Sec. III: The binding energy per Si atom of these two phases on the diboride surface are 1.14 and 1.42 eV, respectively. In agreement with the predictions for preferential sites for the adsorption, planar-like, ($\sqrt{3} \times \sqrt{3}$)-reconstructed silicene is calculated to become the most stable epitaxial structure on ZrB$_2$(0001). When placing the planar-like silicene phase onto the surface, the original phonon instability mentioned for free-standing, planar-like silicene in Sec. III is balanced by the gain of energy of the Si atoms above hollow sites such that the planar-like structure becomes stable against out-of-plane vibrations of those atoms. The second, metastable phase is identified as a deformed version of regularly buckled silicene, where atoms of the top sublattice are in near-bridge sites and those of the bottom sublattice in hollow and on-top sites.28 The on-top Si atom resides at a position higher than those on hollow sites. In the following, we will refer to this structure as the “regularly buckled-like” structure.

For both epitaxial structures, the total binding energy per Si atom defined as $(E[\text{ZrB}_2\ \text{slab}] + E[\text{two silicene layers}] - E[\text{ZrB}_2\ \text{slab}\ +\ \text{two silicene layers}])/12$ is just a little bit larger than 1.0 eV and therefore less than one-third of the binding energy calculated for single Si atoms adsorbed on the same surface. This suggests that the bonding between Si and Zr atoms is weakened by the strong bonding within the silicene layer. However, the bonding of silicene to the diboride surface is still strong enough to significantly impact the electronic structure of the silicene-substrate interface.

The band structures of the regularly buckled-like and planar-like epitaxial phases are shown in Figs. 7(c) and 7(d),

![FIG. 7.](Color online) In-plane and out-of-plane structural parameters of (a) regularly buckled-like silicene and (b) planar-like silicene on ZrB$_2$(0001). The relative heights of Si atoms to the outermost Zr atom are indicated in brackets. The heights of the Zr atoms with respect to the second outermost Zr layer are I: 3.63 Å, II: 3.47 Å, III: 3.47 Å, and IV: 3.47 Å; and I: 3.55 Å, II: 3.50 Å, III: 3.50 Å, and IV: 3.50 Å. (c) and (d) are the band structures of the phases shown in (a) and (b). The Zr $d$ and B $p$ orbital characters are colored in red and blue, respectively. Only the weight at the outermost ZrB$_2$ layer is shown in the fat-band representation. The first Brillouin zone of bulk ZrB$_2$ is adopted.
FIG. 8. (Color online) (a) The (1 × 1) unit cells of ZrB\textsubscript{2} and epitaxial silicene and the corresponding first Brillouin zones. The silicon contributions to the band structures of (b) regularly buckled-like and (c) planar-like silicene on ZrB\textsubscript{2}(0001), represented in the first Brillouin zone of (1 × 1) silicene. The Si \textit{p}\textsubscript{z} and Si \textit{s} orbital characters are colored in black and magenta, respectively, and the Si \textit{p}\textsubscript{x} and \textit{p}\textsubscript{y} characters are colored in green in the fat-band representation. The spectral weight by summing over the orbital contributions of an eigenstate is not shown here.

respectively. Along the Γ-\textit{M} direction of the ZrB\textsubscript{2}(0001) surface, hybridization between Zr\textit{d} and Si \textit{p}\textsubscript{z} orbitals leads to a splitting of the original Zr-derived surface state, resulting in the formation of occupied bonding and unoccupied antibonding bands that have parabolic dispersions resembling that of the bare ZrB\textsubscript{2}(0001) surface. These bands are best identifiable in the planar-like phase. For both phases, B \textit{p}-derived electronic states appear to be almost unaffected by the adsorption of silicene.

While for the discussion of substrate-derived electronic states, the use of the ZrB\textsubscript{2} Brillouin zone is appropriate, silicene-derived states are best discussed using the silicene

FIG. 9. (Color online) The energy barrier between the regularly buckled-like (on the left) and the planar-like (on the right) phases on the diboride surface. The distances of transition images are measured from the regularly buckled-like configuration. The top view and a side view of selected images are presented.
(1×1) Brillouin zone. The relation between the two is illustrated in Fig. 8(a). This would then facilitate comparison with the results for free-standing silicene. As shown in Figs. 8(b) and 8(c), for both phases, at the silicene (1×1) K_{Si} point and in the vicinity of E_F, X-shaped dispersing bands are now missing. This is very different from the case of free-standing silicene, in which cone-like dispersions with p_z character can easily be identified. However, for the regularly buckled-like structure, several bands with partial p_z character are still found below E_F [black color in Fig. 8(b)]. For the one closest to the Fermi energy, in particular, the Si p_z contributions are dominant. This is consistent with the experimental observation of upward two curved features that approach the Fermi level by up to 250 meV at the K_{Si} point that have been observed recently by angle-resolved photoelectron spectroscopy (ARPES). On the other hand, for the planar-like phase, two upward dispersing bands cut through E_F in the vicinity of the K_{Si} point. These bands have only a minor p_z contribution and represent σ bands that are folded from the Γ point as a consequence of the breaking of the translational symmetry. These features do not resemble the features in the ARPES spectra. However, it may be conceived that the parabolic conduction band shifted below the Fermi level following the adsorption of K atoms located halfway between the Γ and K points of silicene (at the M point of the (2×2)-reconstructed ZrB_{2}(0001) surface) bares resemblance to the p_z derived conduction band appearing in the calculations. Instead, a flat π band is found at about 1.0 eV below E_F that carries a larger Si p_z weight. It is obvious that the changes in the electronic structures occurring in particular for the planar-like phase upon adsorption on the diboride surface are not only consequence of the (√3 × √3) reconstruction but to a large degree also caused by interactions with the substrate.

In order to understand if these two epitaxial silicene phases on the ZrB_{2}(0001) surface can be transformed into each other, molecular dynamics calculations have been performed as well. Regardless of whether the initial structure is close to that of the regularly buckled-like form or not, the calculations always converge to the planar-like phase corresponding to the ground state. This indicates that the energy barrier from the regularly buckled-like phase to the planar-like phase is low and has been confirmed by the NEB calculation to be about 15 meV per Si atom in contrast to the deep well at the planar-like configuration. Selected images of transition states and the total energy along the minimum energy path are plotted in Fig. 9. Here, the total energy of the planar-like phase has been set to zero.

**VII. DISCUSSION AND SUMMARY**

Given that a purely sp^2-hybridized, planar free-standing silicene is unstable, it is of great interest to understand the mechanisms that stabilize Si honeycomb lattice configurations when placed epitaxially on surfaces. Along this line, we have investigated planar-like and regularly buckled-like (√3 × √3)-reconstructed silicene phases whose occurrence has been predicted for epitaxial silicene systems. Additionally, these phases have been compared to their respective free-standing, nonreconstructed forms. For a large range of the in-plane lattice constant, free-standing, (√3 × √3)-reconstructed, planar-like silicene is less stable than the regularly buckled form in terms of the total energy. Since it is characterized by one Si atom protruding out of the plane defined by the other five Si atoms of the unit cell, the electronic structure is found to reflect the mixed sp^2/sp^3 bonding nature of this phase. Due to the broken symmetry of the reconstructed surface, the degeneracy of electronic states at the Dirac point of unreconstructed silicene is lifted.

However, with restraining the surface density to exactly six atoms per (2×2) unit cell, the planar-like phase has been found to be the ground state on the Zr-terminated ZrB_{2}(0001) surface. This is because the positions of individual Si atoms adopted in this phase are stabilized by interactions with Zr d orbitals. However, as outlined recently, the experimental data are more consistent with the presence of the metastable phase resembling the regularly buckled form of free-standing silicene and with rather weak interactions at the silicene-diboride interface. In particular, the height profile of the Si honeycomb structure measured by STM does not show any corrugations higher than 0.5 Å. Photoelectron diffraction data are consistent with a structure model in which two of the six Si atoms of the (√3 × √3) unit cell in the so-called “A” positions are surrounded by three “B” atoms at a higher position. Recent ARPES spectra indicate that the parabolic diboride surface state and a silicene-derived band hybridize only after electron donation to the surface following the adsorption of potassium atoms.

On the other hand, epitaxial silicene on the ZrB_{2}(0001) surface is characterized by the spontaneous formation of stress domains, indicating the presence of long-range interactions within the two-dimensional layer of the Si ad-atoms. Having the discussion of the theoretical results presented in this paper in mind, the available experimental data suggest that epitaxial strain weakens the interactions with the substrate such that the stability of possible silicene phases on substrates can be reversed. In order to comprehend the impact of epitaxial strain, it may then be necessary to perform large-scale calculations covering an extended parameter set, which correctly include the effect of interactions between the experimentally observed domains. Nevertheless, it is interesting to note that the ground state of silicene on ZrB_{2} revealed in our DFT calculations, the (√3 × √3)-reconstructed planar-like phase, has also been calculated to form on the Ag(111) surface. Compared to that of free-standing silicene, the phase diagram of epitaxial silicene may be thus much richer.

**ACKNOWLEDGMENTS**

This work was supported by the Strategic Programs for Innovative Research (SPIRE), MEXT, the Computational Materials Science Initiative (CMSI), and Materials Design through Computics: Complex Correlation and Non-Equilibrium Dynamics, A Grant in Aid for Scientific Research on Innovative Areas, MEXT, Japan. This work was also supported by the Funding Program for Next Generation World-Leading Researchers (GR046). The calculations were performed using the Cray XC30 machine at Japan Advanced Institute of Science and Technology (JAIST).